



Synthesis and Electrochemical Properties of $\text{Fe}_2\text{O}_3@\text{C}$ Composite

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Abstract: $\text{Fe}_2\text{O}_3@\text{C}$ material was prepared by one-step hydrothermal method for use as a negative electrode in an iron-air battery. The structure of $\text{Fe}_2\text{O}_3@\text{C}$ was determined by X-ray diffraction (XRD) measurement while their morphology was observed by scanning electron microscopy (SEM). The electrochemical properties of the $\text{Fe}_2\text{O}_3@\text{C}$ electrode in alkaline solution were investigated using cyclic voltammetry (CV) measurement. The results showed that $\text{Fe}_2\text{O}_3@\text{C}$ material with $\alpha\text{-Fe}_2\text{O}_3$ structure and amorphous carbon were successfully synthesized by one-step hydrothermal method. CV measurements indicate that the redox reaction rate of the $\text{Fe}_2\text{O}_3@\text{C}$ electrode is higher than that of the $\text{Fe}_2\text{O}_3@\text{AB}$ electrode using commercial Fe_2O_3 and AB (Acetylene Black Carbon).

Keywords: $\text{Fe}_2\text{O}_3@\text{C}$ material, $\text{Fe}_2\text{O}_3@\text{C}$ electrode, hydrothermal method, iron-air battery.

1. Introduction

The demand for energy storage devices (batteries, supercapacitors...) has been increased rapidly due to their high energy density, long life, reasonable price [1-10]. Previous literatures have shown that metal/air batteries have higher theoretical energy density and specific energy but cheaper, safer than Lithium-ion batteries [7, 11-14]. However, the actual power density of this battery is still low. Therefore, metal/air batteries have been studying to increase their actual cycle performance and capacity. In metal/air battery, the metal is used as the negative electrode material contained in the battery and the oxygen is the positive electrode material that is dispersed into the battery from the air. Most metal/air batteries use aqueous electrolyte such as potassium hydroxide.

Among the metal/air batteries, iron/air batteries have received much attention due to their high theoretical energy, long life, high electrochemical stability, low cost and environmentally friendly [15]. However,

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iron/air batteries still have some limitations such as the instability of iron in the alkaline environment, the passive layer of $\text{Fe}(\text{OH})_2$ formed during discharge and the evolution of hydrogen on the electrodes need to overcome before commercializing.

Our previous work has shown that carbon used as an additive for iron electrodes can increase its cyclability [17]. To overcome the shortcomings of the iron/air battery, $\text{Fe}_2\text{O}_3@\text{C}$ powder was prepared by one-step hydrothermal method and used as electrode material in the iron/air battery to improve its cyclability and capacity.

2. Experimental

Mixture of 0.01 mol of $\text{FeCl}_3 \cdot 6\text{H}_2\text{O}$ (China) dissolved 30ml of deionized water was slowly added into 15ml NaOH solution 2M to obtain a solution containing yellow-brown precipitation. The precipitates thus obtained were washed with distilled water several times to remove Cl^- and Na^+ ions. Add 40ml of 2.5 M NaOH solution and 2.7 g of glucose to the precipitates and this mixture was stirred for 30 minutes, then keep at 160°C in 20 hours using autoclave. After hydrothermal process, the resulting yellow-brown solid was collected by filtration, washed with distilled water or alcohol several times. Subsequently, the product was dried at 60°C for 24h. The obtained compound was identified to be $\text{Fe}_2\text{O}_3@\text{C}$ by X-ray diffraction (XRD). The morphology of the as-prepared $\text{Fe}_2\text{O}_3@\text{C}$ powder was observed scanning electron microscopy (SEM).

To determine the electrochemical behavior of as-prepared $\text{Fe}_2\text{O}_3@\text{C}$, an electrode sheet was prepared by mixing 90 wt.% of the respective $\text{Fe}_2\text{O}_3@\text{C}$ and 10 wt.% polytetrafluoroethylene (PTFE; Daikin Co.) and rolling. The electrodes were cut from electrode sheet into pellets with diameters of 1 cm. The electrode pellets were then pressed onto current collector Ti mesh with a pressure of about 150 kg cm^{-2} .

The $\text{Fe}_2\text{O}_3/\text{AB}$ electrode sheet was prepared by the same procedure with the mixing ratio of $\text{Fe}_2\text{O}_3:\text{AB}:\text{PTFE} = 45:45:10 \text{ wt. \%}$ ($\text{Fe}_2\text{O}_3/\text{AB}:\text{PTFE}=90:10 \text{ wt. \%}$) using Acetylene black (AB) of Denki Kagaku Co. Ltd. and Fe_2O_3 of Aldrich. $\text{Fe}_2\text{O}_3/\text{AB}$ electrodes were made into a pellet of 1 cm diameter.

Cyclic voltammetry (CV) studies were carried out in a three-electrode glass cell assembly that had the synthesized material electrode as the working electrode, Pt mesh as the counter-electrode, and Hg/HgO as the reference electrode. The electrolyte was 8 mol dm^{-3} KOH aqueous solution. CV measurements were taken at a scan rate of 5 mV s^{-1} and within a range of -1.3 V to -0.1 V . In all electrochemical measurements, we used fresh electrodes without pre-cycling.

3. Results and discussion

Structure and morphology of as-prepared material

Figure 1 shows the XRD pattern of as-prepared material. The most typical peaks are characterized by (012), (104), (110), (113), (024), (116), (018), (214) and (300), corresponding to the values of 2θ (degree) at about 24.17, 33.19, 35.66, 40.90, 49.51, 54.13, 57.67, 62.49, and 64.05° respectively in the XRD diagram. They are characterized for a typical pattern of the Fe_2O_3 (ICSD No.82137). Thus, the as-prepared material is Fe_2O_3 . No identifiable XRD signals related to carbon (ICSD No. 1079) are observed. This may be due to the carbon formed in the hydrothermal process has amorphous structure resulting in un-observable the diffraction peaks. To identify the formation of carbon, the SEM measurement was carried out and the result is shown in Fig. 2.

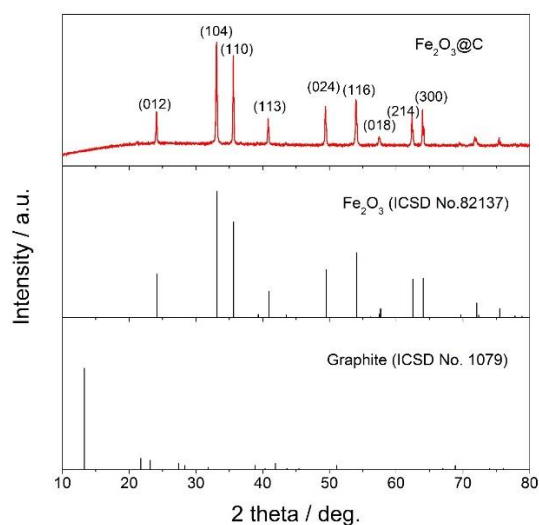


Figure 1. XRD pattern of the $\text{Fe}_2\text{O}_3@\text{C}$

It is clear that the particles with different shapes are covered by thin porous layers. The porous layers that surround the particles are carbon formed during the hydrothermal process while the inner particles are Fe_2O_3 . These Fe_2O_3 particles have micrometer scale in size and un-uniform. The SEM measurement shows that $\text{Fe}_2\text{O}_3@\text{C}$ with α - Fe_2O_3 structure and amorphous carbon were synthesized by hydrothermal method. From these XRD and SEM measurements, it can be concluded that the $\text{Fe}_2\text{O}_3@\text{C}$ material was successfully fabricated by a one-step hydrothermal process.

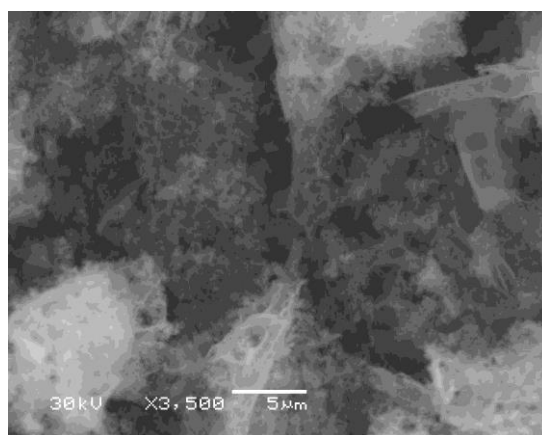


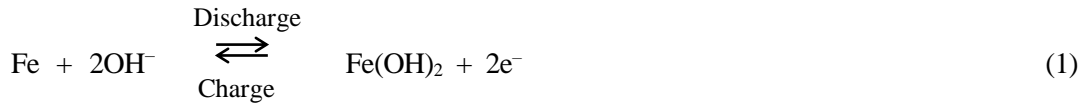
Figure 1. SEM image $\text{Fe}_2\text{O}_3@\text{C}$.

Electrochemical properties

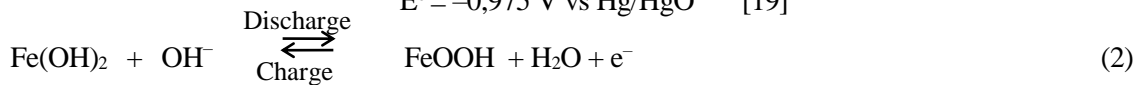
To evaluate the quality of $\text{Fe}_2\text{O}_3@\text{C}$ material synthesized by one-step hydrothermal process, CV measurement was performed at 5 initial cycles (notation 1,2,3,4 and 5) and the results are shown in Fig. 3.

On the forward scan from -1.3 V to -0.1 V, two small oxidation peaks were observed around -1.0 V (a_0) and -0.9 V (a_1) while one small reduction peak occurred around -1.1 V (c_1) together with hydrogen evolution at around -1.2 V on the backward scan.

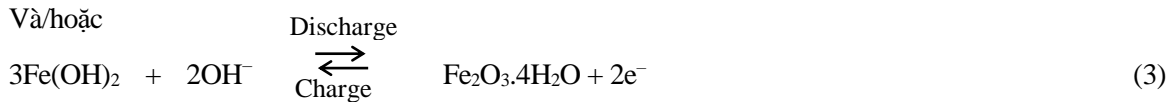
The previous investigation [18] indicated that the clear surface of iron was never exposed to the electrolyte, and over a partially oxidized surface, adsorption of hydroxyl ion takes place. The dissolution of the oxide or underlying metal by the ion transport through the oxide can also take place. The electrochemical reactions of iron in alkaline solution have been reported earlier as the following:



$$E^0 = -0,975 \text{ V vs Hg/HgO} \quad [19]$$



$$E^0 = -0,658 \text{ V vs. Hg/HgO} \quad [19]$$



$$E^0 = -0,758 \text{ V vs. Hg/HgO} \quad [18,20]$$

The first anodic peak a_0 can be attributed to oxidation of iron to $[\text{Fe(OH)}]_{\text{ads}}$, whereas the second anodic peak a_1 can be attributed to oxidation of $[\text{Fe(OH)}]_{\text{ads}}$ to Fe(OH)_2 . The cathodic peak c_1 corresponds to the reduction of Fe(II) to Fe (Eqn. 1). Thus, a_1 and c_1 corresponds Fe/Fe(II) redox couple (Eqn. 1). The redox couple of Fe(II)/Fe(III) (Eqn. 2 and/or 3) was not observable. This could be ascribed to the insulating nature of the Fe(OH)_2 active material, which formed at a_1 peak would inhibit the Fe/Fe(II) redox couple, causing a large over potential.

However, the redox peaks a_1 , c_1 are small, indicating that the redox reaction rate of Fe/Fe(II) (Equation 1) is very slow. This may be due to the porous carbon layer, which surrounds the iron oxide particles prevents the oxidation of iron, leading to slower reaction rate, reducing the cyclability of $\text{Fe}_2\text{O}_3@C$.

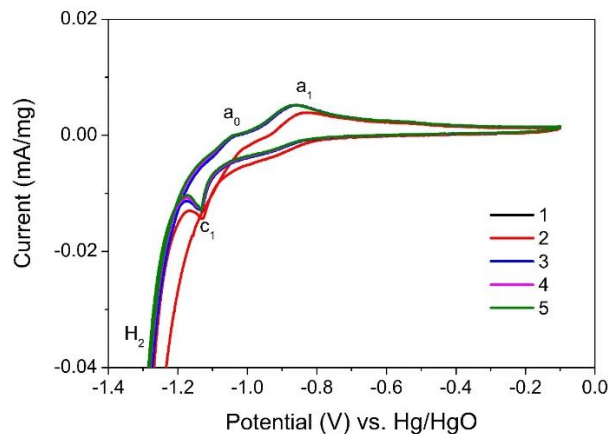


Figure 2. Cyclic voltammetry of $\text{Fe}_2\text{O}_3@C$ electrode with $\text{Fe}_2\text{O}_3@C:\text{PTFE} = 90:10$ wt.% in KOH solution

To fully evaluate the applicability of synthesized $\text{Fe}_2\text{O}_3@\text{C}$, we subjected $\text{Fe}_2\text{O}_3/\text{AB}$ electrode using commercial Fe_2O_3 (Aldrich) and AB carbon (Denki Kagaku Co.Ltd.) for CV measurement to compare with $\text{Fe}_2\text{O}_3@\text{C}$. Figure 4 depicts the SEM images of the commercial AB, Fe_2O_3 and $\text{Fe}_2\text{O}_3/\text{AB}$ powder. The CV profiles of the $\text{Fe}_2\text{O}_3/\text{AB}$ electrode are shown in Fig. 5.

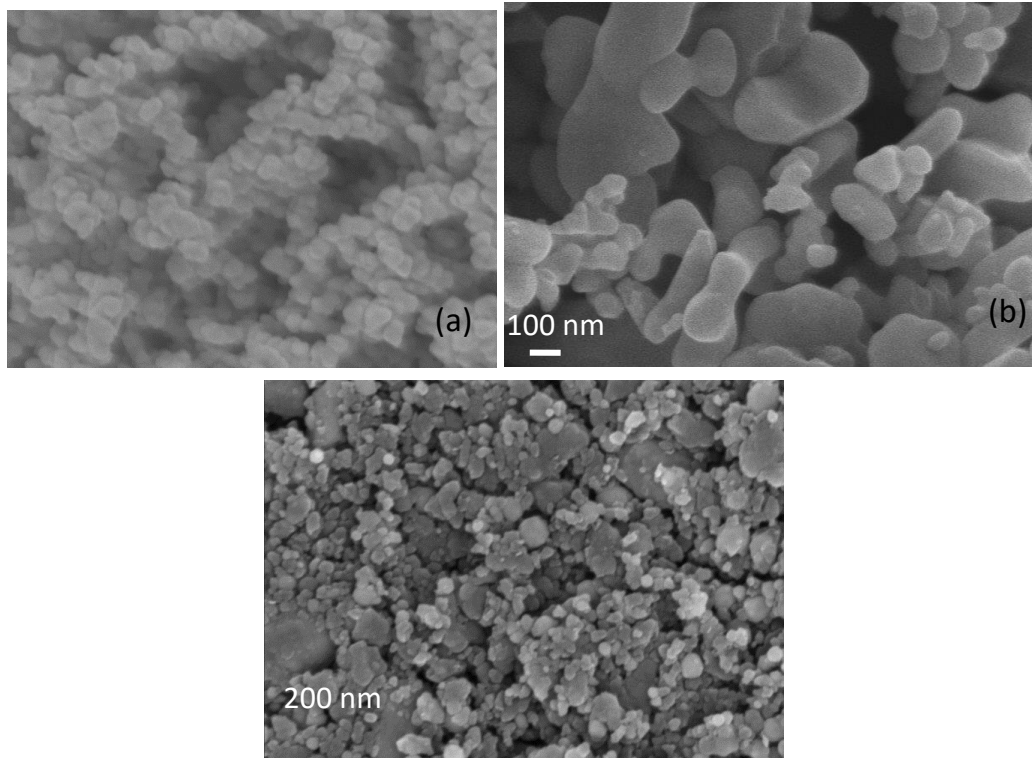


Figure 4. SEM images of commercial (a) AB powder, (b) Fe_2O_3 powder and (c) $\text{Fe}_2\text{O}_3/\text{AB}$ mixture

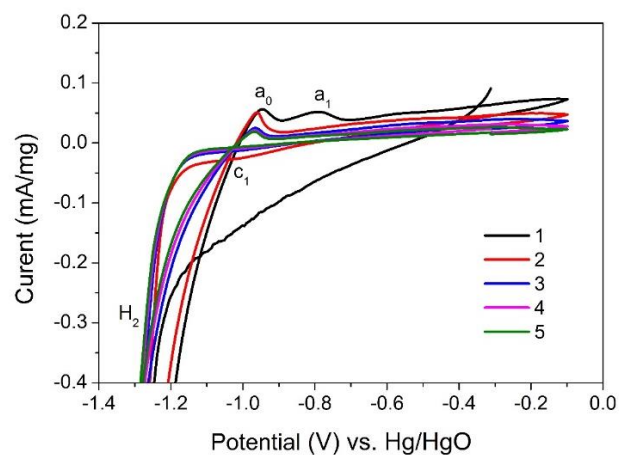


Figure 5. Cyclic voltammetry of $\text{Fe}_2\text{O}_3/\text{AB}$ electrode with $\text{Fe}_2\text{O}_3/\text{AB}:\text{PTFE} = 90:10$ wt.% in KOH solution

SEM image of Fe₂O₃/AB (Fig. 4) is quite different to that of Fe₂O₃@C (Fig. 2). It is impossible to distinguish Fe₂O₃ and AB particles. This suggests that Fe₂O₃ and AB was mixed relative uniformly.

Comparison of the CV results of the Fe₂O₃@C electrode (Fig. 3) with those of Fe₂O₃/AB electrode at corresponding ratio of Fe₂O₃ and AB (Fig. 5) indicates that the redox peaks of Fe₂O₃@C appear more clearly, the reduction peak c₁ is separated from hydrogen evolution while in the Fe₂O₃/AB electrode the redox peaks are lower, the reduction peak c₁ completely covered by hydrogen evolution.

This is a positive behavior of the Fe₂O₃@C material synthesized by hydrothermal process compared to commercial materials. However, the redox current of the Fe₂O₃@C electrode is still low. It may be due to the porous carbon layers surrounded the iron oxide particles inhibit the oxidation of iron, leading to slowdown redox reaction rate. To overcome this phenomenon, the carbon layer formed during the hydrothermal process has to be optimized to increase the cyclability of iron oxide. Consequently, the Fe₂O₃@C material synthesized by this method needs to be further improved to meet the demands for iron-air battery. These steps will be carried out in the subsequent studies.

4. Conclusion

Fe₂O₃@C material has been successfully synthesized by one-step hydrothermal method. Their structure, morphology and electrochemical characteristics were investigated by XRD, SEM and CV measurement. The XRD and SEM results showed that the Fe₂O₃@C material with α -Fe₂O₃ particles covered by amorphous porous carbon was prepared by a simple hydrothermal method, easy to fabricate large amount of material for practical application. Electrochemical measurements indicate that Fe₂O₃@C obtained by hydrothermal process has better cyclability than Fe₂O₃@AB commercial material at corresponding iron oxide and carbon ratio.

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