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Original Article Synthesis and Characterization of a Highly Ordered Mesoporous Bio-glass

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Abstract: A highly ordered mesoporous bio-glass has been successfully prepared by the sol-gel method, in which copolymer pluronic P123 was used as a structure-creating template. The obtained material has the mesoporous structure with the high value of specific surface area ($395.6 \text{ m}^2/\text{g}$) and the 2D hexagonal pore architecture with the pore sizes from 5.5 to 7 nm. The ''in vitro'' experiment was effectuated by soaking the bio-glass powder in the simulated body fluid (SBF). The obtained results confirmed the bioactivity of the synthetic biomaterial through the quick formation of a hydroxyapatite layer after 1 day of immersion.

Keywords: Bio-glass, pore size, mesoporous, bioactivity, "in vitro".

1. Introduction

Bio-glasses are a group of surface-reactive materials that can form a hydroxyapatite (HA) layer when they are soaked in physiological medium or implanted in the human body [1]. Hydroxyapatite phase is the main component of bone mineral, so it acts as an active link between artificial materials and bone tissue, through which the bone defects can be repaired and replaced [2]. The ability to form HA mineral, also called the bioactivity is an important property of bio-glasses and can be controlled by the "in vitro" test in the simulated body fluid (SBF) or by the "in vivo" experiment in the animal body [2,3]. When the HA layer grows as a function of times, the bio-glasses dissolve completely and a new bone matrix is formed without any residue of the starting biomaterials at the end [2-3]. There are different bio-glasses, which have been studied and developed in the past few decades such as bio-glasses 45S5, 46S6,

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58S, 70S30C, S53P4, etc. [1-3]. Two main processes can synthesize these biomaterials, including melting and sol-gel methods. The melting method can synthesize bio-glasses in a short time, about several hours, by heating the starting precursors at high temperatures, following a special regime [4]. Although the melting technique is a quick method, the obtained glasses usually have low values of specific surface area. According to the previous studies, the value of the specific surface area is the key factor, influencing the bioactivity of the bio-glasses [5,6]. The increase in specific surface area can enhance the surface reactions between the artificial materials and the physiological environment, thereby increasing the formation of the HA layer. The sol-gel method can synthesize the bio-glasses at lower temperatures. Especially, the obtained bioglasses can have porous structures and high values of specific surface area that improve the bioactivity of the synthetic materials [7]. To enhance the porous structure and specific surface area of the bio-glasses synthesized by sol-gel method, Pluronic P123-a non-ionic block poly(ethylene copolymer of oxide)-poly (propylene oxide) - poly(ethylene oxide) $[HO(CH_2CH_2O)_{20}]$ (CH₂CH(CH₃)O)₇₀ (CH₂CH₂O)₂₀H] has often been used as a structure-creating template [8-10]. The selfassembly property of this organic compound leads to the highly ordered mesostructure of the synthetic bio-glasses. Depending on the composition of the synthetic bio-glasses and the added amount of the structure-directing agent, the obtained bio-glasses have different properties of pore size, pore-volume, and specific surface area, which influenced the bioactivity ability of the final bio-glasses [9-10]. In this work, the bio-glass 58S with the composition of 58SiO₂-33CaO-9P₂O₅ (wt.%) has been synthesized by the modified sol-gel method. The highly ordered mesostructure of synthetic bio-glass was obtained by using Pluronic P123 as a structure-creating template. The template P123 was added in the synthetic bio-glass with a mass ratio of 1/1. The properties of the obtained bio-glass such as structure morphology, chemical phase and "in vitro" bioactivity have been investigated.

2. Materials and Methods

2.1. Materials

The main chemicals used to prepare the highly ordered mesoporous bio-glass included tetraethyl orthosilicate (TEOS) (reagent grade, 98%, Sigma-Aldrich); triethyl phosphate (TEP) (reagent grade, 99.8%, Merck); calcium nitrate tetrahydrate Ca(NO₃)₂.4H₂O (regent grade, 99%, Sigma-Aldrich) and poly (ethylene oxide)-poly (propylene oxide)-poly (ethylene oxide) (Pluronic P123, chemical formula as HO (CH₂CH₂O)₂₀(CH₂CH(CH₃)O)₇₀(CH₂CH₂O)₂₀H, molecular weight of 5800 g/mol).

2.2. Preparation of highly ordered mesoporous bio-glass

The highly ordered mesoporous bio-glass with the composition of 58SiO₂-33CaO-9P₂O₅ (wt. %) was synthesized with the amounts of reactants presented in Table 1. The chemical precursors were originally calculated to synthesize 2 grams of bio-glass material. Based on the scientific references [11-13], the synthetic processing is briefly described as follows. The mixture of calculated amounts of TEOS and TEP in distilled water was stirred for 30 minutes with the stirring speed of 100 rpm. The nitric acid 2M HNO₃ was used to adjusting the pH of the reaction mixture as 1.5. After that, the calcium nitrate tetrahydrate was added and the reaction mixture kept stirring for 30 minutes. At the end of this time, a clear sol was formed. The amount of P123 was dissolved in the above sol. The mixture was kept in 3 days at room temperature to form the gel. The gel was aged for 2 days at 60°C and then dried at 100°C for 1 day. The dried gel was treated at 700°C for 3 hours to combust the P123 template and convert into the glass powder.

Sample	Nominal composition (%wt)	TEOS	TEP	Ca(NO ₃) ₂ .4H ₂ O	H_2O	P123
		(g)	(g)	(g)	(g)	(g)
58S	58SiO ₂ -33CaO-9P ₂ O ₅	4.03	0.47	2.81	14.62	2

Table 1. Nominal composition (wt. %) and amounts of reactants (g) of the synthetic bio-glass

2.3. "In vitro" experiment

The bioactivity of the synthetic bio-glass was investigated via the "in vitro" test. The powder of bio-glass sample was soaked in the SBF solution in a ratio of 1/2 (mg/mL). The SBF solution used in this study was prepared in the laboratory according to Kokubo's method [14]. The ionic composition of the SBF solution is similar to that of the blood in the human body. The reactive mixture was stirred at a speed of 50 rpm for 1 to 15 days. At the end of immersed time, the powder was rinsed with distilled water and then dried at 100°C. The obtained powder was used for phase identification.

2.4. Physical-chemical characterization

The values of specific surface area, pore size, and pore volume were identified by the N_2 adsorption-desorption measurement on micromeritics ASAP 2010. The BET (Brunauer Emmett - Teller) method was used to determine the specific surface area. The pore volume and pore size were achieved from the desorption branch of the isotherm by the BJH (Barret - Joyner - Halenda) method. For morphological analysis of glass samples, scanning electron microscopy (SEM) (JEOL JMS 6301) and field emission - scanning electron microscopy (FE - SEM) (JEOL JMS 7200F) were used to evaluate the surface morphologies; transmission electron microscopy (TEM) (JEOL JEM 2100) was performed to observe the internal structure. To evaluate the phase structure of the synthetic bio-glass before and after "in vitro" experiment, X-ray diffraction (XRD) measurement were obtained from a powder diffractometer (Bruker D8 Advance). The XRD data were acquired in the range of $10-70^{\circ}$ (20) with a scanning speed of 1°/min. The pH and Si, Ca, P concentration behaviors versus immersion times during the in vitro test were identified by using the pH meter and inductively coupled plasma optical emission spectrometry (ICP-OES) (ICP 2060) technique.

3. Results and discussion

3.1. Nitrogen adsorption/desorption analysis

Nitrogen adsorption/desorption isotherm of the synthetic sample was determined as seen in Figure 1. It could be mentioned that the curve of the sample exhibits the type IV isotherm in the range of relative pressure P/P^0 from 0.7 to 1, which highlighted the mesoporous structure of the synthetic bio-glass. On the other hand, type H₁ of the hysteresis loop is characteristic of opened cylindrical pores in the structure of synthetic bio-glass. The BJH analysis showed the pore size distribution of the synthetic bioglass centered at 6.5 nm as seen in Figure 2. The specific surface area (S_{BET}) and pore volume value (V_p) of the synthetic bio-glass were 395.6 (m^2/g) and 0.52 (cm^2/g) , which are significantly higher than those achieved for conventional solgel bio-glasses of similar compositions [9,10,13].



Figure 1. Nitrogen adsorption/desorption isotherm plot of the synthetic bio-glass.



Figure 2. Pore size distribution curve of the synthetic bio-glass.

3.2. Structural observation

The morphology of the synthetic bio-glass was investigated by using the FE-SEM and TEM observations as seen in Figure 3. The FE-SEM image showed the spherical morphology in the nano-scale, in which the individual particles were agglomerated to form the interconnected porous structure of the obtained material. The TEM analysis highlighted the existence of a long ordered distance with the 2D hexagonal structured porosity. In addition, the pores were quite uniform in size and shape with the pore sizes in the range from 5.5 to 7 nm.



Figure 3. Morphology of the synthetic bio-glass (a) FE-SEM and (b) TEM images.

3.3. Investigation of "in vitro" bioactivity

The bioactivity is the ability to form hydroxyapatite (HA) layer on the glass surface

after "in vitro" experiment at several times. The XRD spectra from 1 day to 15 days were almost similar in appearance. Therefore, the spectrum for 1 day was selected to compare with the one of synthetic bio-glass, as seen in Figure 4. The initial sample showed a broad peak, centered at around 23° (2 θ). This feature of the XRD diagram is characteristic of an amorphous material. The bio-glass sample revealed two well-define crystalline peaks at about 26 and 32° (2θ) after 1 day of "in vitro" experiment. This confirmed the chemical reactions between the bio-glass sample and the SBF environment, leading to the transformation from the amorphous phase to the crystalline phase. By comparison with the XRD standard diagram of the HA material (JCPDS 90432), the two observed peaks are attributed to the formation of HA crystals on the surface of the bio-glass sample. They are corresponding to (002) and (211) crystalline plans in the HA crystalline structure, respectively. The appearance of the new HA phase confirmed the high bioactivity of the synthetic bio-glass. The highly ordered mesoporous structure and high value of specific surface area may be the main factors, enhancing the surface reactions, leading to the quick formation of a hydroxyapatite layer after only 1 day of the "in vitro" test.



Figure 4. XRD patterns of the synthetic bio-glass before and after 1 day of "in vitro" test.

To observe the formation of the HA layer on the surface of the glass sample, the SEM images at different magnifications of the synthetic bioglass before and after 1 day of the "in vitro" test in the SBF solution are presented in Figure 5. The surface of the initial synthetic bio-glass was quite smooth (Figure 5a) while the smalluniform crystals were covered on one of the bioglass after soaking in the SBF solution (Figure 5b1-3). The obtained result confirmed the chemical interaction between the surface of the material and the SBF environment. This leads to forming the HA layer on the surface of the bioglasses [3,4]. In comparison with the previous [4,9,13], the synthetic bio-glass studies expressed the high bioactivity via a quick formation of a new-apatite layer after only 24 hours of the "in vitro" experiment.



Figure 5. SEM observation of the synthetic bio-glass (a)-initial sample and (b1, b2, b3)-samples at different magnifications after 24 hours of immersion in the SBF solution.

3.4. Behaviors of ionic exchange between bioglass and SBF solution

Figure 6 presents the ionic behaviors in the SBF solution as a function of immersion time, identified by the ICP-OES method. According to the scientific papers, the ionic changes in the physiological media are related to the surface reactions between the bio-glass and the SBF solution [3,8,9,13].



Figure 6. Ionic concentrations of the SBF solution during "in vitro" experiment.

For pH, a significant increase was observed during the first five days of immersion. This observation corresponds to the quick ion exchange of Ca^{2+} out of the bio-glass and H⁺ in the SBF solution as the following reaction:

 $(-O-Si-O-)Ca^{2+} + H^{+} \rightarrow -Si-OH + Ca^{2+} (1)$

The consumption of H^+ leads to an increase in pH value. After this period, the pH value was almost constant due to the end of reaction 1.

A similar behavior of Si concentration in the SBF solution was observed. It began with a

period of strong increase, and then reached a saturated step. The increase of Si in the SBF solution is related to the breaking of -Si-O-Si-OH to release the soluble silicic Si(OH)₄.

$$-Si-O-Si-OH + H_2O \rightarrow Si(OH)_4$$
 (2)

The pursuit of reaction 2 was the repolymerization of silicic acid to form SiO_2 – rich layer on the surface of the glass sample.

The existence of Ca and P in the SBF solution consists of two sources. The first one is the Ca, P components available in the initial SBF solution. The second one is the Ca, P amounts released by the reaction of glass with SBF solution. According to reaction 1, the Ca concentration in the SBF solution will increase. However, a significant decrease in the Ca component was identified during the whole immersion process. In addition, a similar observation was also noted for P behavior. This phenomenon is due to the consumption of Ca and P to precipitate the apatite layer on the glass surface. The consumption of Ca and P in the first day is most obvious, confirming the high bioactivity of the synthetic bio-glass. After 1 day of "in vitro", the Ca, P concentrations were almost stable while the pH value increased until 5 days. This phenomenon can be explained by the continuous degradation of bio-glass under reaction 1, simultaneously with the association of Ca and P components to form the HA layer on the surface of the glass sample. The obtained result is consistent with the above analysis by the XRD, in which the HA phase was formed after only 1 day of immersion.

4. Conclusion

Copolymer pluronic P123 was used as a structure-creating agent to prepare the ternary bio-glass $58SiO_2 - 33CaO - 9P_2O_5$ (wt.%) by using the sol-gel method. The synthetic bio-glass showed the highly ordered mesoporous morphology with the high value of the specific surface area and the pore sizes in the range from 5.5 to 7nm. "In vitro" test was effectuated by the immersion of bio-glass powder in the SBF solution. The bioactivity of synthetic bio-glass

was confirmed by the quick formation of the hydroxyapatite phase on its surface after only one day of the "in vitro" experiment.

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